

## Effect of nickel doping on the structural optical and magnetic properties of SnO<sub>2</sub> thin film for solar cell and electronics applications

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### Abstract:

To improve the characteristics of SnO<sub>2</sub> thin films for incorporation into solar cells and in spintronics applications, a comprehensive investigation on the fabrication of pure and Nickel (Ni) doped Tin oxide (SnO<sub>2</sub>) thin films with various doping concentrations has been carried out. By using X-ray diffraction (XRD), scanning electron microscopy (SEM), ultraviolet-visible spectrophotometers, and vibrating sample magnetometers, researchers examined the effects of Ni doping at various concentrations on the structural, optical, and magnetic properties of the various synthesized samples of SnO<sub>2</sub> thin films (VSM). SnO<sub>2</sub> displayed tetragonal structure in all of the Ni-doped samples. The crystal is defective due to the fusing of Ni into the SnO<sub>2</sub> lattice, and the existence of additional peaks demonstrates that the nickel domination is clearly visible. Reduction in the optical band gap was observed with increase of Ni doping levels. The vibrating sample magnetometer measurements revealed that the ferromagnetic signal is progressively enhanced with increase in doping concentration.

**Key Words:** SnO<sub>2</sub> Thin Films, Spray Pyrolysis, Ferromagnetism, Nickel, Doping, Solar Cell, Spintronics.

### 1. Introduction

One of the largest difficulties facing humanity is ensuring a long-term energy source for a sustainable global environment. Fossil fuels and our present energy supply are the key things on which we rely. The use of fossil fuels causes environmental damage through air pollution and global warming. To develop solutions for an energy system that is really sustainable, much research is being done. The research focuses on improving technologies and infrastructure for a smooth transition to renewable energy sources, such as nuclear power, solar power, wind power, geothermal power, biomass and bio fuels, and hydropower, as well as increasing production efficiency, transmission, and utilization of the remaining fossil fuels, positive environmental impacts, [1,2]. From the beginning of the invention of nano structured dye-sensitized solar cell (DSSC), many experiments have been done to improve the performance of the solar cell by utilizing solar energy. Small amount of materials and low-cost manufacturing technologies are used in thin film solar cells. The transparent electrode, which is commonly made of transparent conductive oxides (TCO) is the critical part of the thin film solar cell and there has been current significance on developing new semiconductors for flexible thin-film solar cells [3]. Tin oxide (SnO<sub>2</sub>) is one among the most intriguing materials to be investigated today as it is a well-known n-type semiconductor having a wide band gap of 3.6-3.8 eV [4,5] and due to its potential application in transparent conductive electrode for solar cells, a gas sensing material for gas sensors devices, transparent conducting electrodes, photochemical and photoconductive devices in liquid crystal display, gas discharge display, lithium-ion batteries, etc., [6-9]. Solar cells based on Tin demonstrates optical properties which are superior and good efficiency solar cell materials are exhibited by its alloys [10] having high optical transparency in the visible region and high electrical conductivity which are considered to be the most essential features for solar cells applications, gas sensors, and liquid crystal displays [11]. In addition to these properties SnO<sub>2</sub>

exhibits ferromagnetism and finds applications in novel magnetic optoelectronic devices. Doping of SnO<sub>2</sub> with metal ions has been used to alter the properties of the base material which results in an improvement of the device performance [12]. The specific area of SnO<sub>2</sub> has been increased due to the addition of dopants such as Pt., Pd., Co, Fe and Ni. Of these dopants, nickel has been widely employed to enhance the performance of tin oxide [13]. Due to their close ionic radii: Ni<sup>2+</sup> = 69 nm and Sn<sup>2+</sup> = 71 nm nickel ions can be easily doped into the tin sites. Depending on the doping level there occurs decrease in electron density and increase in oxygen vacancies [14, 15]. SnO<sub>2</sub>: Ni the transparent conducting oxide semiconductor is a direct band gap energy material with nontoxic constituent materials and having abundance in nature. As not much work has been reported on Ni doped SnO<sub>2</sub> semiconductors as per the authors knowledge, an attempt has been made to synthesis Ni doped SnO<sub>2</sub> thin films and to study the influence of Nickel doping on structural, optical and magnetic properties.

## 2. Experimental methods

### Synthesis of Ni doped SnO<sub>2</sub> thin films

Deposition of undoped and Ni-doped (SnO<sub>2</sub>) films on glass substrates at calcined temperature of (500°C) was made by self-assembled spray pyrolysis technique. A homogeneous solution was prepared by dissolving (SnCl<sub>4</sub>.5H<sub>2</sub>O) and (NiCl<sub>2</sub>.5H<sub>2</sub>O) powders in distilled water at the concentration of 0.1 M at room temperature in which the volumetric ratios of Ni were (0, 1, 2, 3 and 4) %. The resultant solution was sprayed on glass substrate. Substrates made of glass and having dimensions 2.5 x 7.5cm<sup>2</sup> (Microscopic glass slides Lab tech medico (P) Ltd, Mumbai) was used for deposition of SnO<sub>2</sub>:Ni thin films. To remove moistures and other organic substance present on the substrate surface, Substrates were washed with distilled water and HCl for several times. Other deposition conditions such as syringe needle substrate distance (8cm), spray time (5s), spray interval (10s), Film thickness (500nm) are employed.

### Characterization Techniques

Scanning electron microscope (HITACHI S-4500H) was employed to obtain SEM images. The structural properties of the prepared SnO<sub>2</sub> thin films were examined by X-ray diffractometer (SHIMADZU-XRD 6000 diffractometer system with X-ray (Cu-K $\alpha$ ) of wavelength 1.5406Å). Unico 4802.UV-Vis double beam spectrophotometer in the wavelength range of 300nm–1100nm was used to measure the optical absorption and transparency of the films. FTIR spectra were recorded by means of SHIMADZU1800-UV Fourier transform infrared spectrometer. Magnetic study was made using a Vibrating Sample Magnetometer (VSM 7410).

## 3. Result and Discussion

### Structural Properties

X-ray diffraction patterns were recorded for Pure and Ni doped SnO<sub>2</sub> thin films using Cu K $\alpha$  radiation of wavelength 1.54 Å. Fig. 1 shows the diffraction peaks obtained for the SnO<sub>2</sub> and Ni doped SnO<sub>2</sub> thin films. The highest diffraction peaks are (110), (101), (200), (211), (220) and (301). The coincidence of diffracted peaks with tetragonal structure of SnO<sub>2</sub> (JCPDS card number: 41-1445) is observed for all the recorded peaks of Ni doped SnO<sub>2</sub> thin films. Ni doping causes no change in the tetragonal structure of SnO<sub>2</sub>. No change in the position of the peak has been observed in XRD spectra which indicates that there is no considerable change in the structure. It is confirmed that nickel was doped into the host SnO<sub>2</sub> lattice as no other peaks related to nickel and nickel oxides were identified. The fusion of Nickel into the SnO<sub>2</sub> lattice makes imperfection in the crystal for 4at% of the dopant concentration and the presence of additional peak confirms that the nickel domination is well observed which is evident from the SEM image and the diffraction pattern. Though the doping level of Ni was increased from 1at% to 3 at% Nickel oxide phases were not dominated in Ni doped SnO<sub>2</sub> thin films. The intensity of the peak corresponding to the predominant plane (110) in the primary phase is increased on Ni doping until 3 at % of doping concentration and further addition of 1

at% doping concentration the intensity decreases due to the replacement of  $\text{Sn}^{4+}$  ions with Ni ions in the lattice of  $\text{SnO}_2$  film, whereas the intensity of the plane (211) in the secondary phase rapidly increases. Which shows better atomic arrangement and lower scattering in these planes. The crystallite size of pure  $\text{SnO}_2$  and Ni doped  $\text{SnO}_2$  thin films for all the dopant levels was calculated using Debye-Scherrer formula [16]

$$D = \frac{0.9\lambda}{\beta \cos\theta}$$

Where  $k$  is a constant,  $\lambda$  is the diffraction wavelength of  $\text{CuK}\alpha$  ( $\lambda = 1.5406 \text{ \AA}$ ),  $\beta$  is the full width at half maximum (FWHM), and  $\theta$  is the diffracted angle, respectively and it is found to be 45nm, 40nm, 35nm, 30nm and 30.5nm.

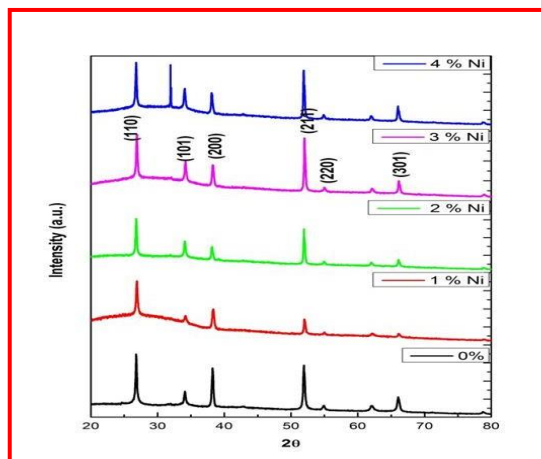


Fig.1. X-ray diffraction spectra of pure  $\text{SnO}_2$  and Ni- doped  $\text{SnO}_2$  thin films.

### Morphological Analysis

Fig.2. shows the variation in the surface morphology of  $\text{SnO}_2$ : Ni thin films deposited at a calcined temperature of  $500^\circ\text{C}$  with different Nickel content concentration. It can be seen in Fig.2.that there is a significant change in the morphology of the films. The presence of uniform sized grains are seen in the undoped film. The evolution in the structure of the films as a function of the dopant concentration is evident from the SEM images. The presence of the dopant Nickel and the quantity that goes into each sample influences the change in microstructure. 1at% of Nickel doping influences the change in the morphology of  $\text{SnO}_2$  film. In the recorded image of 2 at% of Ni doped  $\text{SnO}_2$  thin films grain size decreases and there occurs agglomeration at certain places due to attractive forces between Ni doped  $\text{SnO}_2$  Nanoparticles. Thin film of good surface morphology with a smooth surface having smaller grain size is observed for 3 at % of Ni doped  $\text{SnO}_2$  thin films. Above 3 at % of doping concentration, lot of large irregular grains with different shapes have been identified as Nickel domination is more effective which is also evident from the XRD pattern.

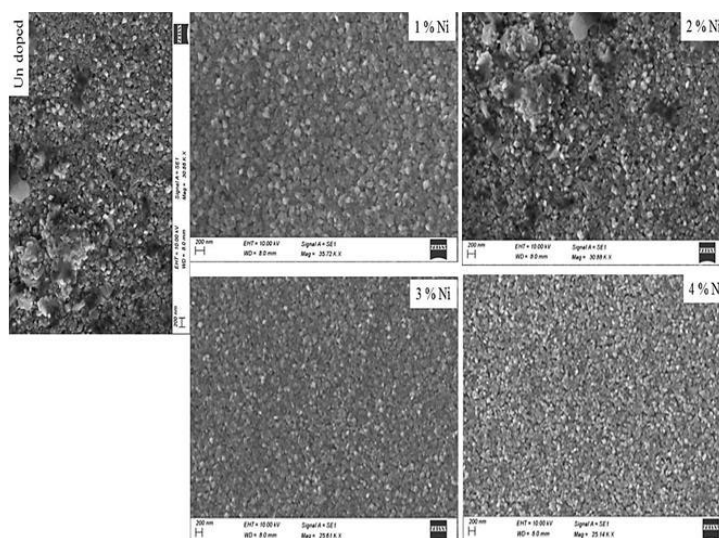


Fig.2. SEM images of pure and Ni doped SnO<sub>2</sub> thin films.

### Optical properties

Optical transmittance spectra of undoped and Nickel doped tin oxide thin films are shown in fig. 3. It can be seen that transparency in the visible region (400–700 nm) is about ~30–90%. Minimum Transmittance is observed for undoped SnO<sub>2</sub> thin films. Due to homogenous and smooth surface as evident from the SEM image, maximum transmittance is observed for 3 at% Ni doped SnO<sub>2</sub> thin film indicating a good quality thin film for disposing it to be used as an electrode material in solar cell. According to Beena et. al., [17] the decrease of transmittance arises due to many factors like increase in thickness, existence of oxygen vacancies and defects, roughness of the surface, permeable nature of the films, scattering in grain boundary etc. but in the present case the decrease of transmittance may be due to the irregular grains.

The absorbance spectra of pure SnO<sub>2</sub> and Ni doped SnO<sub>2</sub> thin films at different concentrations are depicted in fig.4. The recorded spectra shows an ultraviolet cut-off around 300nm–400nm which may be due to the photo-excitation of electrons from valence band to conduction band. High absorption value is observed in the UV region and it becomes low at visible region. The observed absorbance is almost constant over the higher wavelength region. The band gap is an important property which determines the quality of the deposited thin films. Fig.5 displays  $(\alpha h\nu)^2$  vs  $h\nu$  using Tauc's plot for determining the band gap energy. The band gap values ranges from 3.46 to 3.65eV. Kuppamet. al., [18] reported the energy band gap of Nickel doped Tin oxide lies between 3.70eV and 3.76eV and this is contrary to our present findings. In our previous results the band gap energy of pure SnO<sub>2</sub> deposited at a calcined temperature of 500<sup>0</sup>C was found to be 3.5eV [19] which is in close agreement with the value obtained by Pan et al [20] is found to be 3.42eV and it is contrary to our present findings (3.65eV) for undoped polycrystalline SnO<sub>2</sub> thin films. Doping concentrations of Nickel Causes shrinking in band gap energies due to Moss-Burstein effect. In the present case good quality film was obtained in 3at% of Nickel doping concentration with the band gap energy of 3.43eV.

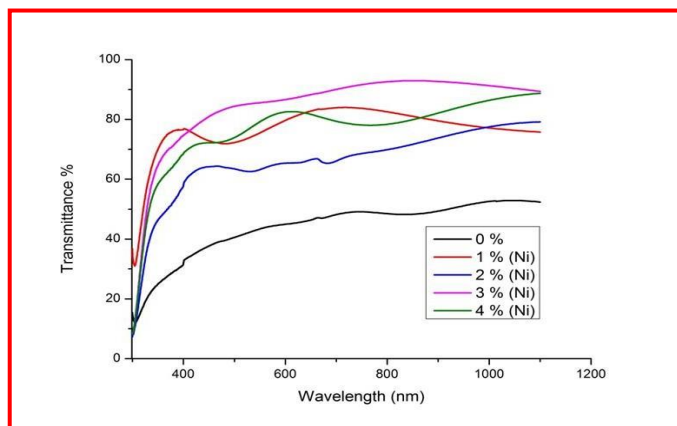


Fig.3. Transmittance spectra of pure SnO<sub>2</sub> and Ni- doped SnO<sub>2</sub> thin films.

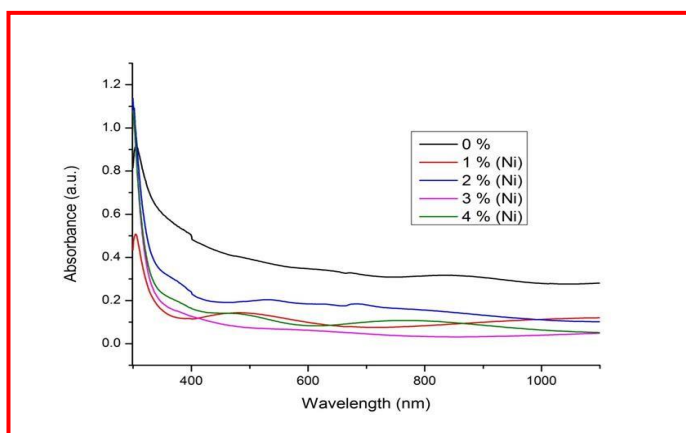


Fig.4. Absorption spectra of pure SnO<sub>2</sub> and Ni- doped SnO<sub>2</sub> thin films.

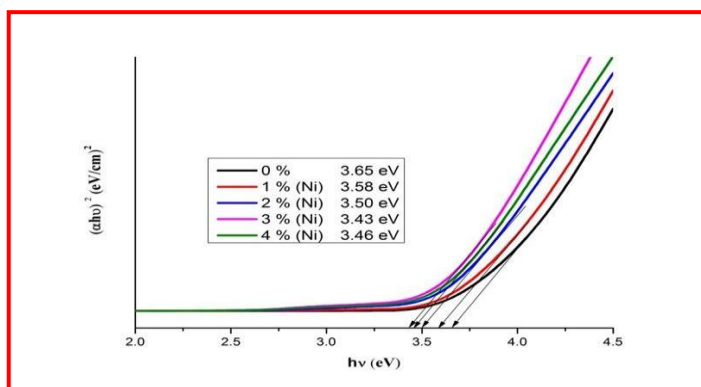


Fig.5. Tauc's plot for pure SnO<sub>2</sub> and Ni doped SnO<sub>2</sub> thin films.

### FTIR analysis

Fig.6. shows the FTIR spectra in the range 500cm<sup>-1</sup> to 4000cm<sup>-1</sup> of pure SnO<sub>2</sub> and Ni doped SnO<sub>2</sub> deposited by self-assembled spray pyrolysis technique. Usually the range between 300cm<sup>-1</sup> and 800cm<sup>-1</sup> have been detected as Sn-O stretching vibration [21]. The peak appearing around 490cm<sup>-1</sup> seems to be O-Sn-O and Ni bond vibration. The band located at 1620cm<sup>-1</sup> is due to H-O-H vibrating mode of the absorbed water. Peaks observed at 1050cm<sup>-1</sup>, and 1320cm<sup>-1</sup> may be assigned to stretching vibrations of C-O and C-H bond. The band 1900 cm<sup>-1</sup> is attributed to Sn-OH vibrational mode.

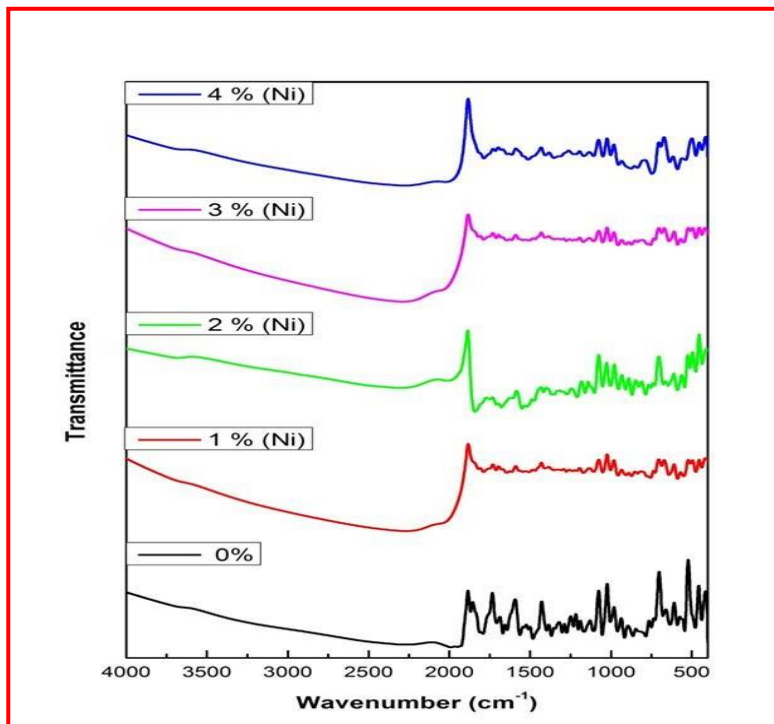


Fig.6. FTIR spectra of pure SnO<sub>2</sub> and Ni- doped SnO<sub>2</sub> thin films.

### Magnetic properties

Fig.7. shows the curves of magnetization versus magnetic field of the pure and Nickel doped SnO<sub>2</sub> samples. The nanoparticles of SnO<sub>2</sub> are diamagnetic in nature and this was confirmed by several groups [22]. While Hays et al. found that nanoparticles of SnO<sub>2</sub> are non-ferromagnetic [23]. In the present case pure SnO<sub>2</sub> exhibits diamagnetic property. Doping of Ni results in the change of magnetization as the property transforms from diamagnetic to ferromagnetic state due to substitution of Ni ion with SnO<sub>2</sub> matrix. The ferromagnetic signal is progressively enhanced upto 3at % of dopant. The saturation magnetization ( $M_s$ ), coercivity ( $H_{ci}$ ) and retentivity ( $M_r$ ) were summarized in the table.1. From the table it is observed that for 3at % dopant level the saturation magnetic moment was high. This may be due to the replacement of Ni<sup>2+</sup> with Sn<sup>4+</sup> leading to the increase of oxygen vacancies available for electron trapping and hence saturation magnetic moment increases. Low retentivity ( $M_r$ ) and coercivity ( $H_{ci}$ ) for 3at% concentration implies that resistance of the ferromagnetic material to become demagnetized is low.

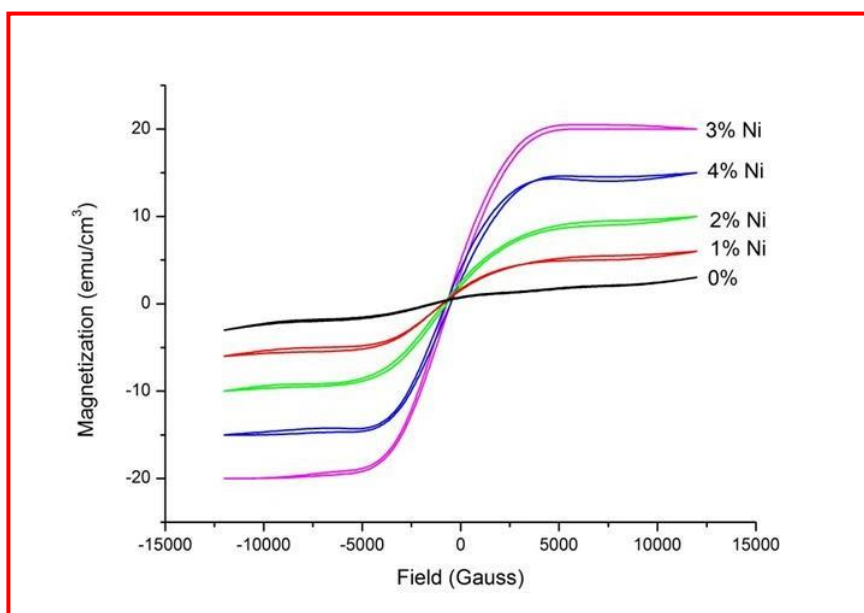


Fig.7. Magnetic property study (M-H curve) at Room temperature for pure and Ni doped SnO<sub>2</sub> samples.

Table.1. Summary of doping concentration, Saturation Magnetisation, Coercivity, Retentivity of SnO<sub>2</sub> thinfilms.

Doping Concentration	M <sub>s</sub> (emu /cm <sup>3</sup> )	H <sub>ci</sub> G	M <sub>r</sub> emu/ cm <sup>3</sup>
0	2.34	188.35	1.09
1at%	3.08	165.22	1.89
2at%	8.62	276.92	2.69
3at%	20.03	96.71	2.16
4at%	15.24	146.37	4.76

#### 4. Conclusion

Using the self-assembled spray pyrolysis process, Ni doped SnO<sub>2</sub> thin films with doping concentrations of 1at%, 2at%, 3at%, and 4at% were effectively produced. The grain size was found to be dependent on the dopant concentration based on the surface morphology of the films. Numerous outstanding characteristics were found in our findings, including crystallinity, high transmittance (90%), a band gap value of 3.43 eV, increased ferromagnetic behaviour, low retentivity (M<sub>r</sub>), and low coercivity (H<sub>ci</sub>). Therefore, a possible contender for solar cells and spintronics applications is nickel doped at a 3 at% concentration in thin SnO<sub>2</sub> films.

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